

# Construction of ZnO devices : electric and magnetic properties

H.Tabata<sup>1</sup>, H.Matsui<sup>1</sup>, H.Saeki<sup>1</sup> and S.Masuda<sup>2</sup>

<sup>1</sup>Osaka University, ISIR-Sanken, 8-1 Mihogaoka, Ibaraki, 567-0047, Japan

<sup>2</sup>Conika-Minoruta Co., Takatsuki, Osaka, 569-8503, Japan

**Zinc oxide is one of the wide band gap semiconductors. By using the ZnO-layer as an active channel layer, transparent transistors (ZnO-TFTs) have been constructed. The ratio  $I_{on}/I_{off}$  of ZnO-TFTs fabricated on Si-wafers is about  $10^6$  and the optical transmittance of ZnO-TFTs fabricated on glass is more than 80%. It is also performed that the magnetic properties of ZnO thin films are controlled by V or Co doping under the non-equilibrium condition. Ferromagnetic properties of Co-ZnO films are confirmed by not only SQUID but also MCD and core-level XAS. Relatively weak signal of ferromagnetics suggests that a part of the film shows ferromagnetic character due to the inhomogeneous of the doped exotic cations (V, Co etc.). These results show that it is possible to fabricate a transparent spin-FET that can be operated even in the presence of visible light**

## I. INTRODUCTION

ZnO is one of the 2-6 compound semiconductors with a wide band gap of 3.3eV [1]-[6]. If we can control p-type and n-type conductivity of this material, it is expected that light emitting devices operating in the short-wavelength range, from blue to ultraviolet, such as LED based on GaN devices. Furthermore, ZnO is one of the rare materials which can be well crystallized even on the amorphous substrates such as glasses, SiO<sub>2</sub>/Si and plastics. Using these characteristics, it is expected that transparent Thin Film Transistors (TFTs) using ZnO as the active channel layer. It is expected that the characteristics of ZnO-TFTs will not degrade on exposure to visible light due to the wide band gap of its active channel layer, whereas the characteristics of amorphous Si TFTs and poly Si TFTs do degrade. Therefore, there is no need to shield the active channel layer from visible light. This makes the TFT structurally-simple and transparent to visible light. This

can allow the aperture ratio of active matrix arrays to be increased.

Here, we have described the construction and characteristics of bottom-gate-type TFTs using ZnO as the active channel layer and a two-layer gate insulator, and discuss the reduction in the leakage current, the drain current characteristics, and the transparent TFT [7].

It is important to control both the electric properties and the magnetic properties of Zn O thin for the practical applications in opto-electronic and opto-magnetic devices at room temperature. We have also fabricated transparent magnetic materials with oxide semiconductors, such as V and Co substituted ZnO films.

## II. EXPERIMENTAL

ZnO thin films were deposited by a pulsed laser deposition using ArF excimer laser which can deposit high quality oxide thin films. The target was a single crystal ZnO substrate (99.9999%) which is the highest-purity oxide available at present. Because of its broad application, an amorphous glass (Corning #1737), SiO<sub>2</sub>/Si and sapphire were used as the substrates. The ZnO thin films of about 250nm thickness were deposited on the substrate at a temperature of 300-550°C in an oxygen atmosphere. The oxygen gas pressure ( $P_{O_2}$ ) was changed from  $6 \times 10^{-4}$  to  $1 \times 10^{-1}$  Torr. For constructing the TFTs, SiO<sub>2</sub> and SiN are used as gate insulating layer.

Transition metal (TM) substituted for Zn in ZnO ( $Zn_{1-x}TM_xO$  ( $x = 0$  to 0.15)) films were formed on sapphire (0001) substrates. A low-temperature process (non-equilibrium process) such as a PLD is effective to introduce a much larger amount of exotic ions (magnetic ions) into the ZnO films, comparing with a conventional synthesize process for constructing bulk materials. Therefore, low temperature formation enables us to expand the solubility limit. Co and V are used as substituting TM elements up to 15%.

The crystal structure was evaluated by X-ray diffraction (XRD). The resistivity, carrier concentration, and Hall mobility,  $\mu_{\text{Hall}}$ , of the films were measured by the Van der Pauw method using a four point probe. The surface morphology of the films was observed using an atomic force microscope (AFM). Magnetic measurements were performed using a superconducting quantum interference device (SQUID) magnetometer with the magnetic field applied parallel to the film plane. The magnetic circular dichroism (MCD) and Vis-UV spectra were measured under the magnetic field of  $-1.5\text{T}\sim+1.5\text{T}$ . XMCD, XAS and PES measurements were also performed under the collaboration of Dr.Fujimori's group.

### III. RESULTS AND DISCUSSION

The 250nm thick  $\text{SiO}_2$  layer had good insulation performance with a leakage current of approximately  $10^{-10}\text{A/cm}^2$  at an applied voltage of  $\pm 5\text{V}$ . However, upon depositing ZnO on the  $\text{SiO}_2$  layer, the leakage current of the ZnO/ $\text{SiO}_2$  double layer increased to  $10^{-4}\text{A/cm}^2$ , and the insulating properties of the  $\text{SiO}_2$  layer were degraded. Even in case of very thick  $\text{SiO}_2$  insulating layer of  $1.2\ \mu\text{m}$ , it is not enough to protect the leakage current up to  $1.0\times 10^{-9}\text{A/cm}^2$ . The double layer gate insulator comprised of  $\text{SiO}_2$  and  $\text{SiN}_x$  is quite effective to realize the low leakage current below  $1.0\times 10^{-11}\text{A/cm}^2$ .

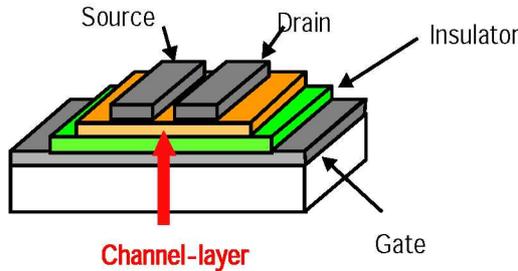


Fig. 1 Schematic model of ZnO-TFT

Using the  $\text{SiO}_2$  and  $\text{SiN}_x$  double gate insulator suppressed the gate-source/drain leakage current and enabled the ZnO-TFT to operate successfully.(Fig.1) Figure 2 shows the electrical characteristics of a ZnO-TFT with a double gate insulator. It shows a typical feature of an enhancement-mode device as there was low drain current below  $10^{-11}\text{A/cm}^2$  at a gate voltage of  $0\text{V}$ . And it shows good transistor character. The ratio of  $I_{\text{on}}/I_{\text{off}}$  was more than  $10^5$  whose value is similar with that of a-Si TFT using in practical device applications [7].

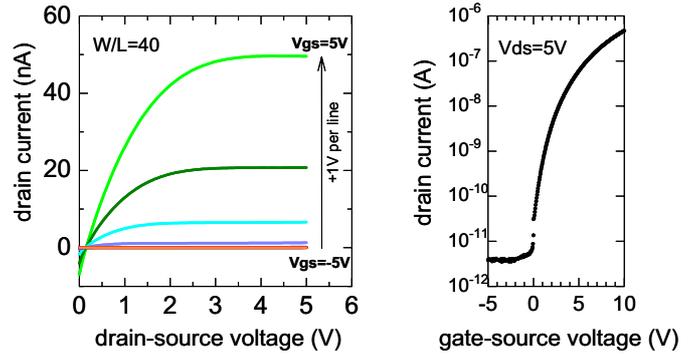


Fig.2  $I_d$ - $V_{ds}$  and  $I_d$ - $V_{gs}$  curves of ZnO-TFT[7]

Theoretical calculations predict the ferromagnetic semiconducting properties should be observed in the transition metal doped ZnO [8]-[10].

Substituted Co and V for Zn have di-valent cationic state in the ZnO thin films formed epitaxially on the sapphire substrates [11]-[12]. XRD pattern and Raman measurements guarantee the wurtzite crystal structure can be maintained up to 15 % and 5% for Co and V substitution, respectively. The results of photo emission spectra also suggest that Co does not have metal state but  $2+$  states. Magnetic circular dichroism(MCD), X-ray MCD and SQUID measurements of (Zn,Co)O films indicate ferromagnetic properties with magnetization of  $0.1\text{-}0.3\ \mu_B/\text{Co}$ . From these results, we can neglect the ferromagnetic properties coming from metal-Co. M-H hysteresis curves are observed by SQUID measurements (Fig.3). But it is not enough value when we think about the origin of double exchange type ferromagnetics. Additional experiments are required to discuss details before elucidating the mechanism of ferromagnetic properties of (Zn, Co)O films.

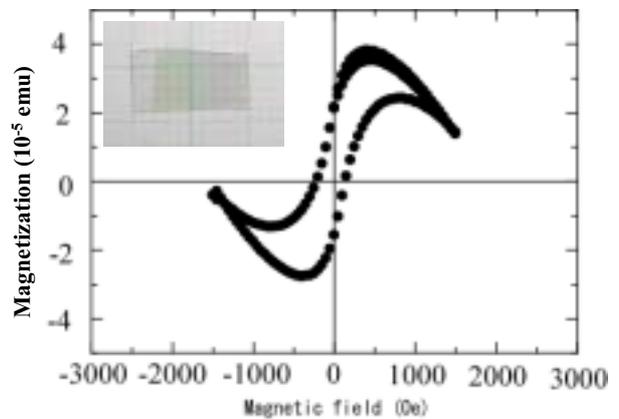


Fig. 3 M-H curve of (Zn,Co)O films. Inset shows photograph of transparent magnetic film

#### IV. CONCLUSION

We have demonstrated the bottom-gate-type TFTs using ZnO as the active channel layer and a two-layer gate insulator. And also ferromagnetic semiconductors are fabricated transition metal doped ZnO films. ZnO is one of the promising wide gap semiconductors having the various kinds of functions.

#### ACKNOWLEDGEMENT

The authors wish to acknowledge the collaboration of Mr. Kobayashi, Mr. Ishida and Prof. Fujimori for measuring the XMCD, XAS and PES spectra.

#### REFERENCES

- [1] H. J. Ko, Y. F. Chen, T. Yao, K. Miyajima, A. Yamamoto and T. Goto: *Appl. Phys. Lett.* 77, 537 (2000)
- [2] M. A. L. Johnson, S. Fujita, W. H. Rowland, W. C. Hughes, J. W. Cook, and J. F. Schetzina: *J. Electron. Mater.* 25, 855 (1996).
- [3] R. D. Vispute, V. Talyansky, S. Choopun, R. P. Sharma, T. Venkatesan, M. He, X. Tang, J. B. Halpern, M. G. Spencer, Y. X. Li, L. G. Salamancariba, A. A. Iliadis, and K. A. Jones: *Appl. Phys. Lett.* 73, 348 (1998).
- [4] Z. K. Tang, G. K. L. Wang, P. Yu, M. Kawasaki, A. Ohtomo, H. Koinuma, and Segawa: *Appl. Phys. Lett.* 72, 3270 (1998).
- [5] D. M. Bagnall, Y. F. Chan, Z. Zhu, T. Yao, S. Koyama, M. Y. Shen, and T. Goto: *Appl. Phys. Lett.* 70, 2230 (1997).
- [6] S. Hayamizu, H. Tabata, H. Tanaka and T. Kawai: *J. Appl. Phys.* 80, 787 (1996).
- [7] S. Masuda, K. Kitamura, Y. Okumura, S. Miyatake, H. Tabata and T. Kawai, *J. Appl. Phys.* 93, 1624 (2003).
- [8] T. Dietl, H. Ohno, F. Matsukura, J. Cibert and D. Ferrand, *Science* 287, 1019 (2000).
- [9] K. Sato and H. Katayama-Yoshida, *Physica B* 308-310, 904 (2001).
- [10] E. -C. Lee and K. J. Chang, *Phys. Rev. B* 65 085209 (2004)
- [11] K. Ueda, H. Tabata and T. Kawai, *Appl. Phys. Lett.*, 75, 4088 (1999).
- [12] H. Saeiki, H. Tabata and T. Kawai, *Solid State Commun.* 120, 439 (2001).

